

Biofilms as complex fluids

James N. Wilking, Thomas E. Angelini, Agnese Seminara,
Michael P. Brenner, and David A. Weitz

Bacterial biofilms are interface-associated colonies of bacteria embedded in an extracellular matrix that is composed primarily of polymers and proteins. They can be viewed in the context of soft matter physics: the rigid bacteria are analogous to colloids, and the extracellular matrix is a cross-linked polymer gel. This perspective is beneficial for understanding the structure, mechanics, and dynamics of the biofilm. Bacteria regulate the water content of the biofilm by controlling the composition of the extracellular matrix, and thereby controlling the mechanical properties. The mechanics of well-defined soft materials can provide insight into the mechanics of biofilms and, in particular, the viscoelasticity. Furthermore, spatial heterogeneities in gene expression create heterogeneities in polymer and surfactant production. The resulting concentration gradients generate forces within the biofilm that are relevant for biofilm spreading and survival.

Introduction

In natural and industrial environments, nearly every surface with a little moisture and nutrients is colonized by bacteria that can live in sessile, interface-associated aggregates referred to as biofilms.^{1,2} Biofilm formation is marked by the production of an extracellular matrix (ECM), which is composed primarily of polysaccharides³ and proteins.⁴ This matrix provides a scaffolding structure that holds the community of cells together and provides the biofilm with mechanical integrity. Bacterial biofilms play a crucial role in global ecology,^{5,6} can be beneficial for water treatment and waste sequestration,⁷ but are also responsible for many bacteria-related problems, including tooth decay,^{8,9} implant infections,^{10,11} hospital-acquired infections,¹² and fouling of industrial processes.¹³ As such, a general framework for understanding biofilm material properties is essential for both the removal of biofilms and the optimization of biofilm properties.

Biofilms can also be viewed from the perspective of soft condensed matter physics; this provides a valuable, alternative materials-related understanding of their structure and properties. In this picture, biofilms are composites of colloids embedded in a cross-linked polymer gel. The bacterial cells are analogous to colloidal particles, and in the absence of the ECM, a bacterial colony behaves as a colloidal fluid. The polysaccharide polymers in the ECM are cross-linked by proteins and multivalent cations,

and the ECM is analogous to a cross-linked polymer gel. In addition, bacteria within the biofilm produce surfactants in order to communicate with one another; these surfactants also control interfacial properties (see the Shrout et al. article in this issue).^{14,15} The production and assembly of cells, polymer, cross-links, and surfactants result in a structure that is heterogeneous and dynamic. In this article, we discuss the physical role of these bacterial biofilm building blocks and review recent progress toward understanding biofilms as composite complex fluid materials.

Polymer gels have an equilibrium water content that is determined by their composition,¹⁶ and biofilms can control their water content by regulating the composition of their ECM. In this manner, they exert control of their mechanical properties, as the mechanics of most soft materials are set by their water content. We gain insight into the mechanical properties of bacterial biofilms, such as their viscoelasticity, by looking to the mechanical behavior of standard complex fluids such as soft colloids and polymer gels.

Recognizing the components of a biofilm as complex fluids facilitates a new understanding of biofilm dynamics. For example, cells can tune the abundance of polymer or surfactant in the extracellular space in response to environmental cues (see Chai et al. and Renner et al. articles in this issue), locally modulating the material properties of the biofilm, and in this

James N. Wilking, Harvard University, Cambridge, MA 02139, USA; jwilking@seas.harvard.edu
Thomas E. Angelini, University of Florida; t.e.angelini@ufl.edu
Agnese Seminara, Harvard University, Cambridge, MA 02139, USA; Agnese.seminara@gmail.com
Michael P. Brenner, Harvard University, Cambridge, MA 02139, USA; Brenner@seas.harvard.edu
David A. Weitz, Harvard University, Cambridge, MA 02139, USA; weitz@seas.harvard.edu
DOI: 10.1557/mrs.2011.71

manner generating spreading forces. Although bacteria are rigid and hardly deform under external stresses, biofilms are dynamic, active materials with the potential of generating forces and adapting to a changing environment. Spatial heterogeneities in the extracellular chemical concentration, including nutrients, oxygen, or intercellular signaling molecules, can result in corresponding heterogeneities in polymer production, cell proliferation rate, and biosurfactant excretion (see the Chai et al. article). Accordingly, these materials not only provide biofilms with structural integrity, but they also can generate forces that drive dynamic structural changes throughout the biofilm life cycle.

Biofilm structure

Biofilms form at interfaces. They can grow to be tens of microns to several millimeters thick and are composed of micron-scale bacterial cells embedded in a soft ECM (**Figure 1a–c**). Bacterial cells are rigid^{17–19} with well-defined shapes such as spheres or rods that are static on time scales shorter than cell division and relatively unresponsive to external mechanical perturbations; by contrast, mammalian cells are easily deformable, highly dynamic in both shape and mechanical properties, and are responsive to external mechanical stress.²⁰ Most bacteria within a biofilm are sessile and lack flagella or pili,²¹ and there are no known extracellular molecular motors capable of generating forces outside of the cell. Instead, bacteria exercise control of the structure and mechanical properties of the biofilm by regulating the composition of the ECM.

The ECM is composed primarily of complex polysaccharides and proteins. Extracellular polysaccharides are mostly anionic with high molecular weights³ $M \approx 10^5$ – 10^6 and are produced in such sufficiently high concentrations that the polymer chains are entangled. Extracellular proteins serve a variety of functions;

importantly, they often act as cross-linkers.²² Cross-links are also provided by polysaccharides, which can form secondary structures such as helices and local crystalline regions.²³ In addition, polysaccharides can form ionic complexes with multivalent cations²⁴ that serve as cross-links, and biofilms are known to form opportunistically in environments where cross-linking ions are present to provide structural integrity. There is also evidence that biofilms can cleave extracellular polysaccharides²⁵ and disrupt protein cross-linking²⁶ in the late stages of biofilm growth.

A full understanding of biofilm material properties remains elusive. This is primarily because the composition of the ECM in naturally occurring biofilms, which often contain multiple species, is highly variable and poorly understood. However, recent advances in our understanding of the genetics of biofilm formation and the composition of the ECM in common laboratory strains have provided well-defined model systems such as *Bacillus subtilis*,^{27–30} *Escherichia coli*,^{31,32} and *Pseudomonas aeruginosa*³³ that are useful for extending our understanding of biofilm mechanics and material properties.

For example, *B. subtilis* biofilms cultured in the laboratory exhibit a fascinating life cycle with material properties that are dynamic and heterogeneous (see the Chai et al. article). *B. subtilis* biofilms growing on the surface of an agar gel expand radially, and bacteria replicate fastest near the leading edge. In a young, expanding colony, rod-shaped bacteria nearest the leading edge are densely packed with little ECM, and bacteria in the center of the biofilm are encased in the ECM. As growth slows due to nutrient depletion, nearly all bacteria are encased in matrix, and the mature biofilm appears highly wrinkled with large ridges covering the biofilm surface (**Figure 1a**). In the late stages of *B. subtilis* biofilm development, severe starvation occurs, and hundreds of cells pile up in aerial structures known as fruiting bodies that serve as preferential sites for sporulation.³⁴ Interestingly, at this stage, due to a combination of chemistry and surface topography, *B. subtilis* biofilms grown on agar show persistent hydrophobicity.³⁵ In the last stage of *B. subtilis* growth, the bacteria produce D-amino acids that induce the detachment of matrix cross-linking proteins from the cell wall, and the biofilm breaks apart.²⁶ In this manner, the biofilm controls its mechanical properties through genetic control of the ECM and even small molecules.

Biofilm mechanics

Control of water content

The ECM is a cross-linked polymer gel, a material that provides the bacteria with exquisite control over the water content in the biofilm. We can understand a great deal about the behavior of the biofilms by exploiting the analogy to soft matter and the polymer-like behavior of the ECM.

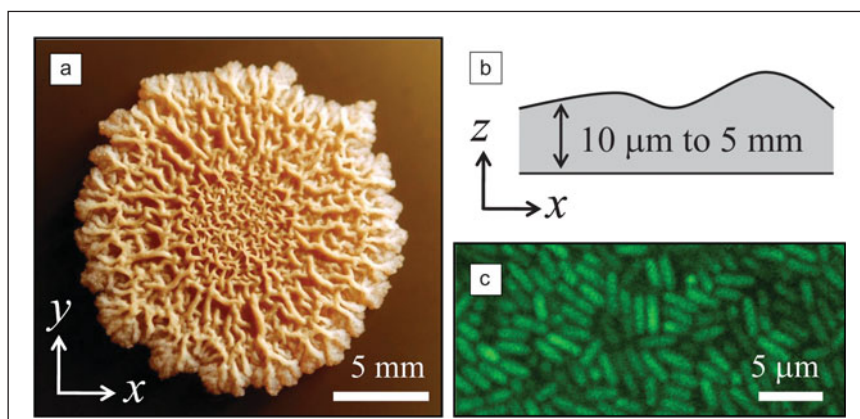


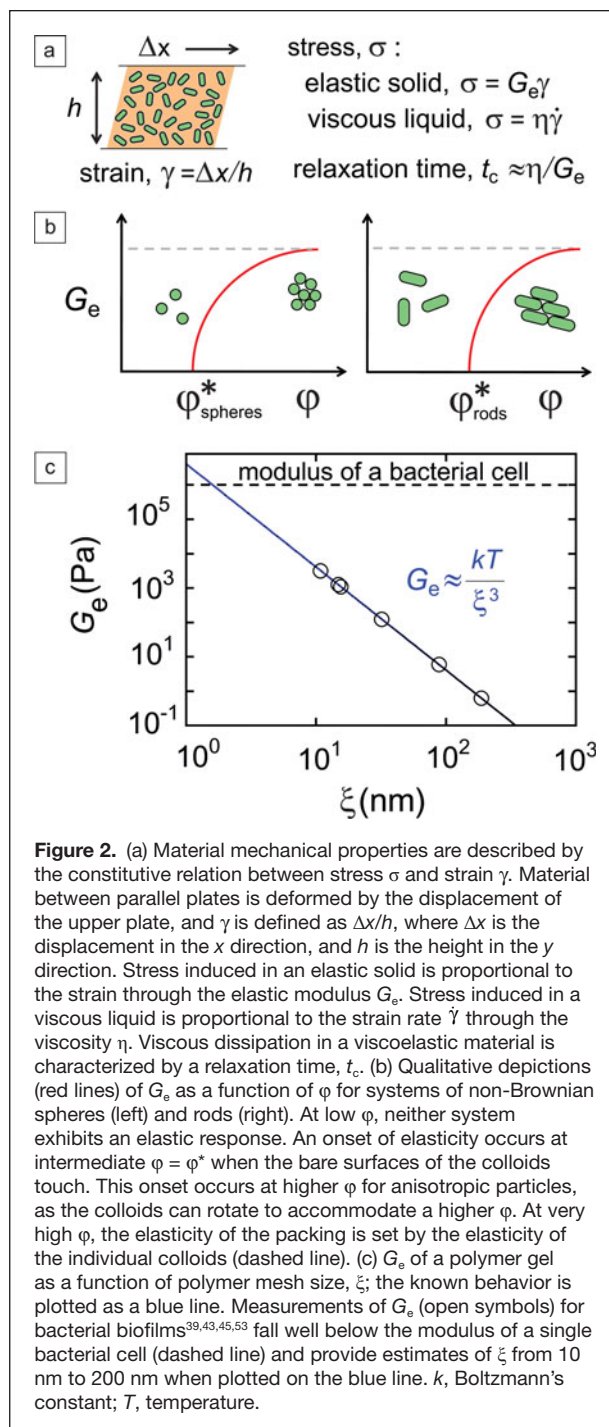
Figure 1. (a) *Bacillus subtilis* biofilm grown on the air-solid surface of agar gel containing water and nutrients. The surface of the agar was inoculated with a drop of planktonic bacteria, and the dish was incubated at 30°C for three days. The biofilm expands radially as it grows, and the mature biofilm exhibits a highly wrinkled morphology. (b) Biofilms exhibit a range in thickness, ten to a thousand times that of a single bacterium. (c) Confocal microscopy image of a *Bacillus subtilis* biofilm. Individual micron-scale bacteria expressing green fluorescent protein are embedded in a viscoelastic extracellular matrix that is not fluorescently stained.

An entangled polymer placed in contact with a reservoir of liquid will swell and disperse throughout the liquid, maximizing entropy by exploring all available configurations in the system. If the polymer and reservoir are separated by a membrane permeable to water but not polymer, the polymer will exert an osmotic pressure, Π , on the membrane. If cross-links are introduced between polymers, as the entangled polymer swells, network strands between cross-links are stretched until Π is balanced by the elastic shear modulus of the gel, G_e .¹⁶ For a given polymer concentration and cross-link density in the ECM, there is an equilibrium water content in the biofilm. For a biofilm in the presence of an abundant water source, such as a pellicle on an air-liquid interface or a submerged biofilm on a liquid-solid interface, cross-links place a limit on the maximum amount of water that can be brought into the biofilm, and in this manner prevent complete dissolution. In the opposite limit, entropic costs resist dehydration of the biofilm. Furthermore, since at equilibrium $\Pi \approx G_e$, the mechanical properties of a gel are directly related to the water content, this provides a means of controlling the mechanics of the biofilm.

Viscoelasticity

Like most soft materials,³⁶ biofilms are viscoelastic.^{37–39} they exhibit a time-dependent response to an imposed mechanical perturbation.⁴⁰ Two relevant measures of material viscoelasticity are the linear elastic shear modulus G_e , which quantifies the force necessary to deform a solid material by a small amount,³⁶ and the relaxation time, t_c of the material, which characterizes the viscous response of the material (**Figure 2a**). Strong biofilm formers are often referred to as “robust,”^{30,41} they are difficult to deform and do not flow rapidly when deformed. Therefore, robustness describes materials having large G_e and long t_c . Insight into the viscoelastic properties of robust biofilms and biofilms, in general, can be gained by considering the mechanics of model viscoelastic materials.

The viscoelastic properties of well-defined systems such as colloidal pastes and polymer gels are uniquely dependent on the fraction of liquid in the material, and the details of this dependence lie in the physical properties and interactions of the colloids or polymer.^{36,42} For soft, micron-scale colloids, the relevant parameter is the colloidal volume fraction, ϕ . A qualitative depiction of $G_e(\phi)$ for a disordered system of repulsive soft spheres is shown in Figure 2b; this behavior is relevant for cocci, bacteria with spherical shapes. There is an onset of elasticity at an intermediate volume fraction ϕ^* , whereas the elastic modulus of the packing at high ϕ is set by the elasticity of the individual colloids. At $\phi < \phi^*$, the material is a liquid suspension, and stress relaxes immediately. At $\phi > \phi^*$, the material is a solid paste that exhibits stress relaxation, which is logarithmic in time, so residual stress can remain in the system for very long times. Interestingly, viscoelastic behavior reminiscent of colloidal pastes has been observed in *Staphylococcus aureus* biofilms, which are composed of densely packed spherical bacteria that lack significant ECM.⁴³ $G_e(\phi)$ for a disordered system of repulsive soft micron-scale rods is also shown in Figure 2b;



this behavior is relevant for bacilli, bacteria with rod-shaped cells. Because the rods can rotate when packed,⁴⁴ the system exhibits an onset of elasticity at ϕ higher than that of spheres.

Although biofilms are 80% or more water by volume, much of this water is contained within the rigid bacterial cells, so it is useful to consider a bacterial volume fraction ϕ_b analogous to the colloidal volume fraction. While the high ϕ behavior of colloidal packings is likely relevant for biofilms with large ϕ_b , such as biofilms grown on the air-gel interface of agar plates or infections in tissue wounds, progress in the fundamental

understanding of biofilm mechanics has been limited by the material and genetic complexity of biofilms, and the mechanics of dense biofilms have not been well characterized. In most biofilms, ϕ_b is low, in particular those in contact with reservoirs of liquid. Confocal microscopy images¹ indicate that most bacteria within biofilms are separated by distances of $\sim 0.5 \mu\text{m}$ or more, so ϕ_b is typically less than 0.2; this value is so low that the elasticity of the biofilm is determined predominantly by the properties of the ECM.

Within the picture of a soft material, the elasticity of a polymer gel arises from entropy; deformation of the gel leads to extension of polymer strands, which is resisted by entropic fluctuations. As a result, the elastic moduli, in units of energy per unit volume, is proportional to thermal energy, kT , where k is Boltzmann's constant and T is temperature, and inversely proportional to a volume. The relevant volume is set by the polymer entanglement mesh size, ξ . Thus, the elastic modulus is $G_e \approx kT/\xi^3$. This known dependence of G_e on ξ is plotted in Figure 2c. Measurements of G_e reported in the literature vary from approximately 1 Pa to 10^3 Pa;^{39,43,45,53} this provides an estimate for ξ of roughly 10 nm to 200 nm, but no known measurements of ξ exist in the literature.

When subjected to large deformations, some biofilms exhibit strain-stiffening,^{45,46} as shown in Figure 3. This behavior is commonly observed in biopolymer gels,⁴⁷ but the physics of strain-stiffening in biofilms is not understood. In the reported measurements, the strain γ is not homogenous throughout the sample, and ϕ_b may also change with γ . If strain-stiffening proves to be a general feature of biofilm mechanics, it may be relevant to the removal of biofilms from surfaces.

Differences in the composition and production of ECM contribute to the broad range in biofilm mechanical properties. In addition, the availability of water in natural and industrial settings appears to contribute to the mechanics of the biofilm. For example, biofilms such as *B. subtilis* on agar gel plates or multi-species infections in tissue wounds grow in direct contact with a viscoelastic material; in the process of imbibing water, they must perform work by dehydrating the surrounding material. As a result, their water content is set by the external osmotic pressure rather than by the equilibrium between entropy and energy; a sobering example of this are *P. aeruginosa* colonies in the concentrated lung mucus of cystic fibrosis patients.⁴⁸ In contrast, biofilms that form on the interior surface of hospital catheters and as pellicles on the surface of standing water are in direct contact with aqueous liquid. These biofilms do not perform work against their environment when imbibing water; as a result, their water content is high. Measurements show that biofilms grown on agar gels have higher elasticities than those grown in direct contact with a water source, but the role of the mechanics of the surrounding material in determining biofilm mechanics remains unexplored.

Measurement techniques

Many of the tools and techniques developed for measuring the mechanics of soft matter systems can be applied, with slight

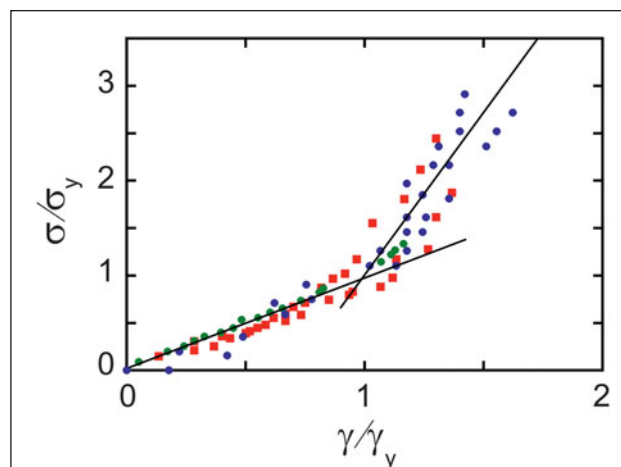


Figure 3. Stress σ as a function of strain γ reported for three different types of bacteria: *Staphylococcus epidermidis* (■) and *Klebsiella pneumoniae* (●) measured using a deformable microfluidic device⁴⁵ and *Pseudomonas aeruginosa* (●) measured by deforming the biofilms with fluid flow.⁴⁶ All three biofilms exhibit strain-stiffening behavior; the characteristic shape is seen when the σ and γ are normalized by the yield stress σ_y and yield strain γ_y , respectively. Lines guide the eye. Strain-stiffening is relevant to the removal of biofilms from surfaces, but the physics of this behavior is unclear.

modifications, to bacterial biofilms. Biofilms are often heterogeneous, fragile, and microscopically thin; thus nontraditional techniques are essential to measure their mechanical properties. For example, passive microrheology⁴⁹ is a powerful technique that is just beginning to be applied to bacterial biofilms.⁴³ This technique relies on the thermally driven motion of tracer colloids to provide local rheological information. However, in living systems such as biofilms, care must be taken in extracting rheological information.^{50,51} Microbead force spectroscopy⁵² has been used to measure the mechanical properties of biofilms grown on the surface of a spherical atomic force microscope tip. In a new microfluidic technique, biofilms are grown to fill the inside of a deformable microfluidic channel, and the biofilm is deformed by applying pressure in an adjacent channel.⁴⁵ Other promising techniques include the use of micropipette cantilevers^{53,54} and interfacial dilation.⁵⁵

Biofilm dynamics

Surfactant heterogeneity and cooperative motility

Bacteria move for many reasons: to invade host tissues, to seek out nutrients, or to erect structures for spore spreading. Moreover, bacteria motility is often cooperative in nature (also see the Shrout et al. article in this issue). For example, classes of cooperative surface motility include swarming, twitching, gliding, social gliding, adventurous gliding, sliding, and spreading.^{56,57} Swarming is a type of cooperative surface-associated motility in which each cell is self-propelled by its own flagella. Twitching and social gliding also depend on a motor-driven cellular appendage; Type IV pili are extended toward a surface, adhered, then retracted, pulling the individual cell forward toward the adhesion. Intriguingly, many types of collective

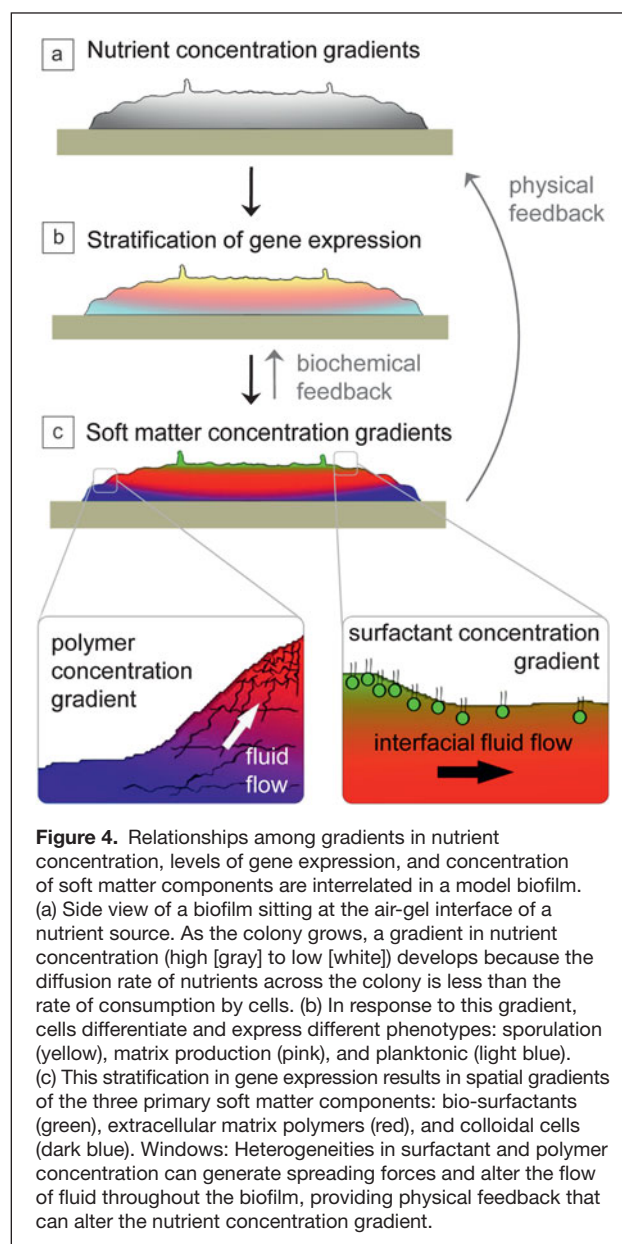
motility do not require the use of known motor-driven appendages such as flagella or Type IV pili; these include gliding, adventurous gliding,⁵⁸ sliding, and spreading.^{59–61} The physical mechanisms underlying these kinds of cooperative motility are not yet clear, yet a clue comes from the physical chemistry of their excretions; all of these motor-independent types of motility involve amphiphilic molecules. These types of motor-independent motility depend on the extracellular production of fatty acids, *N*-acyl-homoserine lactones, neutral lipids, phospholipids, glycolipids, glycopeptidolipids, flavolipids, lipopeptides, proteins, and lipoproteins.^{14,62–66} Complex fluid properties of these amphiphilic molecules are valuable for understanding these types of collective motility.

Spatial heterogeneities in surfactant concentration lead to gradients in interfacial tension, and the resulting Marangoni flows of interfacial fluids can generate spreading forces.⁶⁷ Surface tension gradients in biofilms have not been extensively studied, although it is widely recognized that the surface-active nature of biosurfactants underlies the physical driving force for motility.^{68,69} Frequently, cell locomotion is attributed to the ability of these molecules to reduce interfacial tension and lubricate surfaces. Indeed, collective spreading of bacteria colonies can be controlled by the addition of exogenous surfactant;⁷⁰ however, a mere reduction in interfacial tension or friction does not generate a driving force for motility. By contrast, spatial gradients in surfactant concentration at the surface of a colony or biofilm can generate spreading forces. This type of spreading force has been shown recently in plate-cultured colonies of the soil bacterium *Rhizobium etli*, which forms symbiotic associations with the root nodules of many legumes, and reported for wall-climbing *Bacillus subtilis* biofilms grown on air-fluid interfaces.^{14,15} Remarkably, the Navier-Stokes equations that are used to describe fluid motion quantitatively predict the spreading rate and thickness of the bacterial film,¹⁵ when solved numerically for the surfactant-driven spreading of fluid drops. Moreover, the surfactant-driven spreading of fluid drops exhibits spatial patterns that are highly reminiscent of those observed at the edges of spreading colonies; this suggests that Marangoni flows may be a newly discovered mechanism for bacterial pattern formation, complementary to established mechanisms.^{71,72}

Material heterogeneity and nutrient depletion gradients

Heterogeneity in polymer concentration within a biofilm has major physical consequences; gradients in osmotic pressure and mechanical elasticity can guide fluid flow and localized compression or expansion of material. It is possible to explain such heterogeneities by citing genetic variation throughout the biofilm, yet spatial gradients in cell behavior can arise from a simpler source: metabolite depletion (also see the Shrout et al. article). In general, an aggregate of cells without an internal nutrient transport system can grow in size to an upper limit that is between 100 μm and 1 mm. Growth bottlenecks occur for both eukaryotic and prokaryotic cell aggregates feeding on diffusing nutrients, including tissues and bacterial colonies. This

size limit arises from a depletion of metabolic substrate inside of the aggregate; nutrients and gases that diffuse into the aggregate from the outside are consumed by peripheral cells at the expense of the inner cells. Consequently, a spatial gradient of nutrient concentration develops in this reaction-diffusion system. This nutrient transport limitation has major consequences in tissue engineering and 3D eukaryotic cell culture applications,⁷³ and evidence is emerging that such spatial gradients in cellular metabolites contribute to the physiological functioning of the bacterial biofilms as wholes.⁷⁴ The physical consequences of spatio-temporal heterogeneity in nutrient or oxygen concentration within a biofilm are not yet well understood, but they are expected to be significant. For example, shifts in metabolic activity accompany dramatic increases in ECM production, which can transform the colony from a fluid-like to solid-like material.⁷⁵



In a remarkable example of the biofilm response to nutrient limitations, biofilms in flow cells develop channels that facilitate convective transport of nutrients throughout the colonies, potentially mitigating the nutrient diffusion limitation.⁷⁶ The development of channels within the biofilm has not been directly connected to an internal nutrient depletion response, but the development of structural heterogeneities suggests an association. Spatial gradients in metabolic substrate concentration give rise to a corresponding heterogeneity in cellular metabolic activity; cellular metabolism increases and decreases with the concentration of available metabolic substrate.⁷⁷ Accordingly, the structure and extracellular composition of a biofilm varies as a function of depth within the colony; microelectrode and micro-slicing measurements carried out in combination with confocal microscopy show that the spatial distribution of metabolic substrate concentration and oxygen generates a corresponding spatial distribution of cell density and extracellular pore space.⁷⁸ A similar correspondence between proximity to nutrient and cellular phenotype has been observed in *Bacillus subtilis* biofilms grown on air-agar interfaces (**Figure 4a–b**); cells near the edge of the biofilm, where there is high nutrient level, express motility genes; cells in the bulk of the biofilm, with intermediate levels of nutrient, express high levels of ECM; at the surface exposed to air, the farthest from the nutrient source, cells express early sporulation genes (see the Chai et al. article in this issue).²¹ This genetic stratification is accompanied by corresponding physical properties (**Figure 4c**); we expect the biofilm to behave more as a fluid at the bottom near the edges and as a hydrogel in the middle of the colony. At the top of the colony, where the nutrient concentration is the lowest, spore-filled fruiting bodies form. The physical driving force behind the erection of fruiting bodies is not known; however, their formation depends upon both biosurfactant expression and EPS expression, and starvation can directly induce sporulation.³⁴ Thus, the spatial organization of several genetic phenotypes within biofilms, as well as the cell density and porous structures throughout biofilms, suggest links among local physical properties, gene expression, and the proximity to nutrient or oxygen sources, potentially influencing overall biofilm physiology.

Summary

Bacterial biofilms are complex materials that can be viewed from the perspective of soft matter physics; this perspective provides insight into the structure and dynamics of the biofilm. The mechanics of the biofilm can be understood in light of the primary components: rigid cells embedded in a polymer gel. Furthermore, heterogeneities in the expression of polymer and surfactants in the extracellular matrix set up concentration gradients within the biofilm, which generate forces relevant for spreading and survival. Future advances in our understanding of biofilm genetics and biofilm composition will lead to a clearer picture of the biofilm as a complex fluid. Specifically, a better understanding of the genes responsible for biofilm formation in bacterial infections and industrial fouling will provide the

biological framework necessary for studying these biofilms as materials. In addition, new methods for measuring local compositions of biofilm components will guide in the appropriate application of soft matter physics principles. Advancing our understanding of the biofilm as a complex fluid will benefit the biomedical field and industry by aiding in biofilm removal and preventing biofilm growth.

Acknowledgments

This work was supported by BASF, the NSF (DMR-1006546), and the Harvard MRSEC (DMR-0820484).

References

1. M. Ghannoum, G.A. O'Toole, Eds., *Microbial Biofilms*, (ASM Press, Washington, DC, 2004).
2. G. O'Toole, H.B. Kaplan, R. Kolter, *Annu. Rev. Microbiol.* **54**, 49 (2000).
3. I.W. Sutherland, *Microbiology* **147**, 3 (2001).
4. S.S. Branda, A. Vik, L. Friedman, R. Kolter, *Trends Microbiol.* **13**, 20 (2005).
5. W.B. Whitman, D.C. Coleman, W.J. Wiebe, *Proc. Natl. Acad. Sci. U.S.A.* **95**, 6578 (1998).
6. J.W. Costerton, Z. Lewandowski, D.E. Caldwell, D.R. Korber, H.M. Lappin, *Annu. Rev. Microbiol.* **49**, 711 (1995).
7. B.E. Rittmann, P.L. McCarty, *Environmental Biotechnology: Principles and Applications* (McGraw-Hill Book Co., New York, NY, 2001).
8. P.E. Kolenbrander, *Annu. Rev. Microbiol.* **54**, 413 (2000).
9. P.E. Kolenbrander, R.N. Andersen, D.S. Blehert, P.G. Eglund, J.S. Foster, R.J. Palmer, *Microbiol. Mol. Biol. Rev.* **66**, 486 (2002).
10. A.G. Gristina, *Science* **237**, 1588 (1987).
11. J.L. del Pozo, R. Patel, *Clin. Pharmacol. Ther.* **82**, 204 (2007).
12. K.L. Frank, J.L. del Pozo, R. Patel, *Clin. Microbiol. Rev.* **21**, 111 (2008).
13. J.W. Costerton, K.J. Cheng, G.G. Geesey, T.I. Ladd, J.C. Nickel, M. Dasgupta, T.J. Marrie, *Annu. Rev. Microbiol.* **41**, 435 (1987).
14. R. Daniels, S. Reynaert, H. Hoekstra, C. Verreth, J. Janssens, K. Braeken, M. Fauvart, S. Beullens, C. Heusdens, I. Lambrechts, D.E. De Vos, J. Vanderleyden, J. Vermant, J. Michiels, *Proc. Natl. Acad. Sci.* **103**, 14965 (2006).
15. T.E. Angelini, M. Roper, R. Kolter, D.A. Weitz, M.P. Brenner, *Proc. Natl. Acad. Sci.* **106**, 18109 (2009).
16. M. Rubinstein, R.H. Colby, *Polymer Physics* (Oxford University Press, NY, 2003).
17. A. Cerf, J.C. Cau, C. Vieu, E. Dague, *Langmuir* **25**, 5731 (2009).
18. G. Francius, O. Domenech, M.P. Mingeot-Leclercq, Y.F. Dufrene, *J. Bacteriol.* **190**, 7904 (2008).
19. M. Arnoldi, M. Fritz, E. Bauerlein, M. Radmacher, E. Sackmann, A. Boulbitch, *Phys. Rev. E* **62**, 1034 (2000).
20. K.E. Kasza, A.C. Rowat, J.Y. Liu, T.E. Angelini, C.P. Brangwynne, G.H. Koenderink, D.A. Weitz, *Curr. Opin. Cell Biol.* **19**, 101 (2007).
21. H. Vlamakis, C. Aguilar, R. Losick, R. Kolter, *Genes Dev.* **22**, 945 (2008).
22. D. Romero, C. Aguilar, R. Losick, R. Kolter, *Proc. Natl. Acad. Sci. U.S.A.* **107**, 2230 (2010).
23. E. Lahaye, T. Aubry, N. Kervarec, P. Douzenel, O. Sire, *Biomacromolecules* **8**, 1218 (2007).
24. I. Klapper, J. Dockery, *SIAM Rev.* **52**, 221 (2010).
25. A. Boyd, A.M. Chakrabarty, *Appl. Environ. Microbiol.* **60**, 2355 (1994).
26. I. Kolodkin-Gal, D. Romero, S.G. Cao, J. Clardy, R. Kolter, R. Losick, *Science* **328**, 627 (2010).
27. C. Aguilar, H. Vlamakis, R. Losick, R. Kolter, *Curr. Opin. Microbiol.* **10**, 638 (2007).
28. S.S. Branda, J.E. Gonzalez-Pastor, E. Dervyn, S.D. Ehrlich, R. Losick, R. Kolter, *J. Bacteriol.* **186**, 3970 (2004).
29. S.S. Branda, F. Chu, D.B. Kearns, R. Losick, R. Kolter, *Mol. Microbiol.* **59**, 1229 (2006).
30. D.B. Kearns, F. Chu, S.S. Branda, R. Kolter, R. Losick, *Mol. Microbiol.* **55**, 739 (2005).
31. P.N. Danese, L.A. Pratt, R. Kolter, *J. Bacteriol.* **182**, 3593 (2000).
32. L.A. Pratt, R. Kolter, *Mol. Microbiol.* **30**, 285 (1998).
33. G.A. O'Toole, R. Kolter, *Mol. Microbiol.* **30**, 295 (1998).
34. S.S. Branda, J.E. Gonzalez-Pastor, S. Ben-Yehuda, R. Losick, R. Kolter, *Proc. Natl. Acad. Sci. U.S.A.* **98**, 11621 (2001).
35. A. Epstein, B. Pokroy, A. Seminara, J. Aizenberg, *Proc. Natl. Acad. Sci. U.S.A.* (2010).
36. R.G. Larson, *The Structure and Rheology of Complex Fluids* (Oxford University Press, New York City, 1999).

37. W.G. Characklis, *Biofilm development and destruction. Final report.* (Electric Power Research Institute, Palo Alto, CA, 1979).
38. H.C. Flemming, G. Schaule, *Werkst. Korros.-Mater. Corros.* **45**, 29 (1994).
39. P. Stoodley, Z. Lewandowski, J.D. Boyle, H.M. Lappin-Scott, *Biotechnol. Bioeng.* **65**, 83 (1999).
40. B.F. Piciologlou, N. Zelter, W.G. Characklis, *J. Hydraulics Div., ASCE* **106**, 733 (1980).
41. S.S. Yoon, R.F. Hennigan, G.M. Hilliar, U.A. Ochsner, K. Parvatiyar, M.C. Kamani, H.L. Allen, T.R. DeKievit, P.R. Gardner, U. Schwab, J.J. Rowe, B.H. Iglewski, T.R. McDermott, R.P. Mason, D.J. Wozniak, R.E.W. Hancock, M.R. Parsek, T.L. Noah, R.C. Boucher, D.J. Hassett, *Dev. Cell* **3**, 593 (2002).
42. P.-G. de Gennes, *Scaling Concepts in Polymer Physics* (Cornell University Press, Ithaca, NY, 1979).
43. S.S. Rogers, C. van der Walle, T.A. Waigh, *Langmuir* **24**, 13549 (2008).
44. A. Donev, I. Cisse, D. Sachs, E. Variano, F.H. Stillinger, R. Connelly, S. Torquato, P.M. Chaikin, *Science* **303**, 990 (2004).
45. D.N. Hohn, J.G. Younger, M.J. Solomon, *Langmuir* **25**, 7743 (2009).
46. P. Stoodley, R. Cargo, C.J. Rupp, S. Wilson, I. Klapper, *J. Ind. Microbiol. Biotechnol.* **29**, 361 (2002).
47. C. Storm, J.J. Pastore, F.C. MacKintosh, T.C. Lubensky, P.A. Janmey, *Nature* **435**, 191 (2005).
48. H. Matsui, V.E. Wagner, D.B. Hill, U.E. Schwab, T.D. Rogers, B. Button, R.M. Taylor, R. Superfine, M. Rubinstein, B.H. Iglewski, R.C. Boucher, *Proc. Natl. Acad. Sci. U.S.A.* **103**, 18131 (2006).
49. T.G. Mason, D.A. Weitz, *Phys. Rev. Lett.* **74**, 1250 (1995).
50. T.M. Squires, T.G. Mason, *Annu. Rev. Fluid Mech.* **42**, 413 (2010).
51. D. Mizuno, D.A. Head, F.C. MacKintosh, C.F. Schmidt, *Macromolecules* **41**, 7194 (2008).
52. P.C.Y. Lau, J.R. Dutcher, T.J. Beveridge, J.S. Lam, *Biophys. J.* **96**, 2935 (2009).
53. S. Aggarwal, R.M. Hozalski, *Biofouling* **26**, 479 (2010).
54. S. Aggarwal, E.H. Poppele, R.M. Hozalski, *Biotechnol. Bioeng.* **105**, 924 (2010).
55. B. Klein, P. Bouriat, P. Goulas, R. Grimaud, *Biotechnol. Bioeng.* **105**, 461 (2010).
56. J. Henrichs, *Bacteriol. Rev.* **36**, 478 (1972).
57. R.M. Harshey, *Annu. Rev. Microbiol.* **57**, 249 (2003).
58. M.J. McBride, *Annu. Rev. Microbiol.* **55**, 49 (2001).
59. I.I. Brown, C.C. Hase, *J. Bacteriol.* **183**, 3784 (2001).
60. J. Recht, R. Kolter, *J. Bacteriol.* **183**, 5718 (2001).
61. J. Recht, A. Martinez, S. Torello, R. Kolter, *J. Bacteriol.* **182**, 4348 (2000).
62. T.P. Huang, A.C.L. Wong, *Res. Microbiol.* **158**, 702 (2007).
63. S. Mukherjee, P. Das, R. Sen, *Trends Biotechnol.* **24**, 509 (2006).
64. G. Agusti, O. Astola, E. Rodriguez-Guell, E. Julian, M. Luquin, *J. Bacteriol.* **190**, 6894 (2008).
65. A.A. Bodour, C. Guerrero-Barajas, B.V. Jiorle, M.E. Malcomson, A.K. Paull, A. Somogyi, L.N. Trinh, R.B. Bates, R.M. Maier, *Appl. Environ. Microbiol.* **70**, 114 (2004).
66. C.R. Stewart, O. Rossier, N.P. Cianciotto, *J. Bacteriol.* **191**, 1537 (2009).
67. L.E. Scriven, C.V. Sternling, *Nature* **187**, 186 (1960).
68. R.F. Kinsinger, M.C. Shirk, R. Fall, *J. Bacteriol.* **185**, 5627 (2003).
69. R.F. Kinsinger, D.B. Kearns, M. Hale, R. Fall, *J. Bacteriol.* **187**, 8462 (2005).
70. A. Be'er, R.S. Smith, H.P. Zhang, E.-L. Florin, S.M. Payne, H.L. Swinney, *J. Bacteriol.* **191**, 5758 (2009).
71. O.K. Matar, S.M. Troian, *Phys. Fluids* **11**, 3232 (1999).
72. E. Ben-Jacob, O. Schochet, A. Tenenbaum, I. Cohen, A. Czirok, T. Vicsek, *Nature* **368**, 46 (1994).
73. R. Derda, A. Laromaine, A. Mammoto, S.K.Y. Tang, T. Mammoto, D.E. Ingber, G.M. Whitesides, *Proc. Natl. Acad. Sci. U.S.A.* **106**, 18457 (2009).
74. P.S. Stewart, M.J. Franklin, *Nat. Rev. Microbiol.* **6**, 199 (2008).
75. A.P. White, A.M. Weljie, D. Apel, P. Zhang, R. Shaykhtudinov, H.J. Vogel, M.G. Surette, *PLoS One* **5** (2010).
76. J.D. Seymour, S.L. Codd, E.L. Gjersing, P.S. Stewart, *J. Magn. Reson.* **167**, 322 (2004).
77. S.A. Rani, B. Pitts, H. Beyenal, R.A. Veluchamy, Z. Lewandowski, W.M. Davison, K. Buckingham-Meyer, P.S. Stewart, *J. Bacteriol.* **189**, 4223 (2007).
78. T.C. Zhang, Y.C. Fu, P.L. Bishop, *Water Environ. Res.* **67**, 992 (1995). □



7TH INTERNATIONAL
DENDRIMER
SYMPOSIUM

June 26 - July 1, 2011
Gaithersburg, Maryland, USA

MEETING CHAIR

Anil K. Patri
Nanotechnology Characterization Laboratory
SAIC Frederick, Inc.

www.mrs.org/ids7

REGISTER NOW!

Final Registration Deadline—June 12, 2011

Due to security clearances, on-site registration will not be permitted

Scheduled for June 26 - July 1, 2011, and held at the National Institute of Standards and Technology (NIST) in Gaithersburg, Maryland, the **7th International Dendrimer Symposium (IDS7)** will be this year's prime forum to highlight current research in dendrimers, dendronized polymers and other hyperbranched architectures. In addition to an exceptional scientific and technical program, IDS7 will feature plenary sessions, a combination of invited and contributed talks, poster sessions, and a technology showcase.

The **IDS7 Program** includes the following topics.

- Dendrimer Synthesis and Characterization
- Hyperbranched Polymer Synthesis and Characterization
- Biological Studies
- Drug Delivery
- Imaging
- Computational Modeling and Databases
- Novel Applications
- Catalysis and Light Harvesting
- Device Applications

True to the spirit of the Symposium, a social program with Welcome Reception, Symposium Dinner, and other social functions are being planned to foster participant interaction and collaboration in the historical environment of the Washington, D.C. area. For the most up-to-date information on IDS7, including program information, registration, schedules, travel/hotel arrangements, and more, visit **www.mrs.org/ids7**.



2011



FALL MEETING

November 28-December 2 • Boston, MA



CALL FOR PAPERS

Abstract Deadline • June 21, 2011

MATERIALS RESEARCH SOCIETY
Advancing materials. Improving the quality of life.



SYMPOSIA

ENERGY AND THE ENVIRONMENT

- A Material Challenges in Current and Future Nuclear Technologies
- B Advanced Materials for Fuel Cells
- C *In Situ* Studies of Solid-Oxide Fuel-Cell Materials
- D Sustainable Synthesis of Nanomaterials
- E Advanced Materials for Solar-Fuel Generation
- F Mobile Energy
- G Applications of Hierarchical 3D Structures
- H Organic Photovoltaic Devices and Processing
- I Fundamental Processes of Solar Harvesting in Excitonic Solar Cells
- J Photonic and Plasmonic Materials for Enhanced Photovoltaic Performance
- K Materials for High-Performance Photonics

FUNCTIONAL MATERIALS

- L Topological Insulator Materials
- M Oxide Semiconductors—Defects, Growth, and Device Fabrication
- N Diamond Electronics and Biotechnology—Fundamentals to Applications V
- O Compound Semiconductors for Generating, Emitting, and Manipulating Energy
- P Ferroelectric and Multiferroic Materials
- Q Magnetoelectric Composites
- R Compliant Electronics and Photonics
- S Solution Processing of Inorganic and Hybrid Materials for Electronics and Photonics
- T Large-Area Processing and Patterning for Active Optical and Electronic Devices III
- U Charge Generation/Transport in Organic Semiconductor Materials
- V Multifunctional Polymer-based Materials
- W Phonons in Nanomaterials—Theory, Experiments, and Applications
- Y Advances in Energetic Materials Research

NANOMATERIALS

- Z Functional Metal-Oxide Nanostructures
- AA Carbon Nanotubes, Graphene, and Related Nanostructures
- BB Functional Nanowires and Nanotubes
- CC Functional Semiconductor Nanocrystals and Metal-Hybrid Structures
- DD Transport Properties in Polymer Nanocomposites II
- EE Self Organization and Nanoscale Pattern Formation
- FF Mechanical Nanofabrication, Nanopatterning, and Nanoassembly
- GG Safety and Toxicity Control of Nanomaterials

BIOMATERIALS

- HH Bioelectronics—Materials, Properties, and Applications
- II BioMEMS—Materials and Devices
- JJ Nanofunctional Materials, Nanostructures, and Nanodevices for Cancer Applications
- KK Biomaterials for Tissue Regeneration
- LL Synthetic and Biological Gels
- MM Micro- and Nanoscale Processing of Biomedical Materials
- NN Nucleation and Growth of Biological and Biomimetic Materials
- OO Multiscale Mechanics of Hierarchical Materials

MATERIALS EXPLORATION

- PP Three-Dimensional Tomography of Materials
- QQ Functional Imaging of Materials—Advances in Multifrequency and Multispectral Scanning Probe Microscopy and Analysis
- RR Dynamics in Confined Systems and Functional Interfaces
- SS Properties and Processes at the Nanoscale—Nanomechanics of Material Behavior
- TT Microelectromechanical Systems—Materials and Devices V
- UU Combinatorial and High-throughput Methods in Materials Science

MEETING CHAIRS

Cammy R. Abernathy
University of Florida
cabern@mse.ufl.edu

Paul V. Braun
University of Illinois-Urbana
pbraun@illinois.edu

Masashi Kawasaki
Tohoku University
kawasaki@ap.t.u-tokyo.ac.jp

Kathryn J. Wahl
Naval Research Laboratory
kathryn.wahl@nrl.navy.mil

www.mrs.org/fall2011